

# Synthesis of novel TiO<sub>2</sub> by mechanical alloying and heat treatment-derived nanocomposite of TiO<sub>2</sub> and NiTiO<sub>3</sub>

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A novel nanocomposite titanium dioxide powders were synthesized by mechanical alloying and heat treatment. The absorption threshold of the nanocomposite TiO<sub>2</sub> moved to a longer wavelength and the photoreactivity was dramatically enhanced. Also the new absorption was believed to be induced by the trapping band gap between the valence and conduction bands of nanocomposite TiO<sub>2</sub>.

**KEY WORDS:** titanium dioxide; photocatalyst; mechanical alloying; titanate; nickel.

## 1. Introduction

Titanium dioxide is widely used as a photocatalyst, because of its chemical stability, low cost and harmless nature in the human body. However, TiO<sub>2</sub> becomes active only under ultraviolet light which has an energy greater than the band gap of TiO<sub>2</sub> (~3.0 eV). Overcoming the lack of a reaction in the visible light region is the focus of current TiO<sub>2</sub> research. Therefore, many studies have sought to improve the photocatalytic properties of TiO<sub>2</sub> by doping with cation [1] or anion metals [2] and oxides [3]. However, the photocatalytic properties of the cation-doped TiO<sub>2</sub> decreased even in the UV region. This is because the doped materials have an increase in the carrier-recombination centers and are unstable at high temperature [4]. Also, the absorption spectra of anion-doped TiO<sub>2</sub> in the visible light region are in the comparatively small range of about 400 nm [5].

Recently, Lin *et al.* [6] reported that nickel titanate can decompose the organic compounds by a photocatalytic reaction. Nickel titanate (NiTiO<sub>3</sub>) has attracted a great deal of interest for use in chemical and electric materials due to its weak magnetism and semiconductivity. Conventionally, preparation of NiTiO<sub>3</sub> requires sintering of NiO and TiO<sub>2</sub> at high temperature. The sintering process causes the grain growth which prohibits the production of nanoscaled NiTiO<sub>3</sub> powder [7].

In our previous study, nanocrystalline Ni-doped rutile TiO<sub>2</sub> synthesized by mechanical alloying (MA) and HPPLT (Homogeneous Precipitation Process at Low Temperatures) exhibited an absorption threshold in the range of 480–500 nm [8]. However, this is a very small region in the visible light spectrum. In the current

study, a novel nanocomposite titanium dioxide (Ni-doped TiO<sub>2</sub>+NiTiO<sub>3</sub> by heat treatment) was synthesized by MA and heat treatment. The photocatalytic behaviors of the powders were characterized by measuring the visible light reaction capacity. The detailed microstructural characteristics of the novel nanocomposite TiO<sub>2</sub> and the effect of NiTiO<sub>3</sub> formation was also investigated.

## 2. Experimental

Novel nanocomposite TiO<sub>2</sub> powders were prepared by MA and heat treatment. MA is a very effective process for refining grain size down to the nano-sized range and has the ability to alloy immiscible elements. In order to get a better doping effect, meta-stable rutile powder was selected as the starting powder for MA, instead of a stable TiO<sub>2</sub> phase. To obtain the meta-stable powder, a TiO(OH)<sub>2</sub> precipitate slurry was first prepared from TiOCl<sub>2</sub> using the HPPLT process [9]. Then, the solution was filtered using distilled water. The detailed HPPLT process has been reported elsewhere [9,10]. The filtered precipitates were dried at 60 °C for 12 h to obtain the dried TiO<sub>2</sub> powder. The dried powder was mechanically alloyed for 14 h by a planetary ball mill (Fritz mill, P-5) with nickel powders (KOJUNDO Chem. CO., LTD, 99.9%). The content of the Ni was 8 wt%. The ball-milling speed was 150 rpm and the ball to powder weight ratio was 15:1. In order to control of the impurity effect, the ZrO<sub>2</sub> ball (1 mm diameter) and bowl were used in the ball milling. After milling, the powders were treated at 1000 °C for 4 h in order to obtain the composite powders.

The structural properties of the alloying process and nanocrystallization process including grain size

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determination were characterized by XRD (CuK $\alpha$ , Rigaku D-MAX 3000) and by transmission electron microscopy (TEM, 200 KV, JEM 2000). The characteristics of the visible light reaction were investigated by UV/VIS-DRS (diffuse reflectance spectroscopy, Perkin-Elmer) and photoluminescence (PC-1 Photon counting fluorescence spectrometer). In addition, the synthesized powder was compared to commercial NiTiO<sub>3</sub> (KOJUNDO Chem. CO., LTD, 99.9%) powder and the simple mixture powder of our synthesizing rutile TiO<sub>2</sub> and NiTiO<sub>3</sub> (5 wt%), synthesizing rutile TiO<sub>2</sub> and P-25. Photocatalytic reaction of the powders was estimated by measuring the decomposition rates of 4-chlorophenol (0.1 mmol) in aqueous solution (500 ml) containing 0.1 g of various powders. As the light source a fluorescence lamp (380–800 nm) for visible light and 80 W pressure mercury arc lamp for ultra violet light were used, which was placed at about 20 cm from the quartz beaker. The solution was continuously stirred. During irradiation, the solution was took out at 5, 20, 35 min, each 50 ml. After the photolysis, the remaining carbon concentrations in the aqueous solutions were detected by a total organic carbon analyzer (SHIMADZU 4000 autosampler).

### 3. Results and discussion

Figure 1 shows the X-ray diffraction pattern for P-25, rutile, NiTiO<sub>3</sub>, the alloyed powder (Ni-doped TiO<sub>2</sub>), the alloyed powder after heat treatment and simple mixture of rutile and NiTiO<sub>3</sub> powder, respectively. It can be seen that alloyed powder and rutile powder consisted of only a rutile phase, and the main peak of alloyed powder shifted from the original rutile's main peak. This indicates that there was grain refinement of the average

grain size and an increase in the internal strain by a mechanical deformation during ball milling [11]. Also, P-25 powder is a mixture of rutile and anatase phases. However, after heat treatment, the alloyed powder formed composite of the NiTiO<sub>3</sub> and rutile phase. The weight percent of NiTiO<sub>3</sub> in alloyed powder was calculated to be 5 wt% with the following equation [12]:

$$\text{Content of NiTiO}_3 = \left\{ \frac{I_{\text{NiTiO}_3}}{I_{\text{NiTiO}_3} + 1.265 I_R} \right\} \times 100$$

where  $I_{\text{NiTiO}_3}$  and  $I_R$  are peak intensities of NiTiO<sub>3</sub> and rutile, obtained from X-ray diffraction patterns. It is thought that heat treatment caused a portion of the Ni-doped TiO<sub>2</sub> to form a stable NiTiO<sub>3</sub> phase. However, the peak of the alloyed powder after heat treatment remained broadening and their intensities decreased with heating. It means that the grain growth did not occur through heat treatment because of drag effect [13]. Owing to NiTiO<sub>3</sub> segregation in Ni-doped TiO<sub>2</sub> matrix, grain boundary mobility was controlled by the lattice diffusion of the NiTiO<sub>3</sub> [14,15]. On the other hand, simple mixture powders showed the coexistence of NiTiO<sub>3</sub> and rutile TiO<sub>2</sub> phase. Applying the Debye–Scherrer formula [11] and the full width at half maximum (FWHM) to the (110) reflection, typical values of crystallite sizes have been calculated to be in the range of 5–15 nm for the heat treatment powders. They are in agreement with the sizes determined from TEMs.

The microstructure of the P-25, rutile, NiTiO<sub>3</sub>, alloyed powder and novel nanocomposite powders were examined by TEM. The microstructures are shown in figure 2. In the case of P-25 powders, spherical particles were observed. In addition, the rutile particle had a chestnut bur shape with a size of 200–300 nm, in which the primary acicular particles are coagulated. The NiTiO<sub>3</sub> consisted of conglomerated spherical particles with a grain size of 50–100 nm. The alloyed powder consisted of spherical particles with an average grain size of less than 6 nm. Even though the novel nanocomposite TiO<sub>2</sub> powder had spherical particle, its SAD (selected Area electron Diffraction) pattern was slightly different from the alloyed powder. Except for the novel nanocomposite powder, all of the powders had a SAD pattern that was only one crystal structure. However, the novel nanocomposite TiO<sub>2</sub> powder consisted of nano-sized rutile TiO<sub>2</sub> and rhombohedral NiTiO<sub>3</sub>. Also the particles were uniform with an average grain size less than 15 nm. The small grain size and uniformity was caused by welding and fracture of the powder particle during ball milling. These results agree well with the XRD data in figure 1.

Figure 3 show the results of diffuse reflectance spectra result of P-25, rutile, NiTiO<sub>3</sub> powder, the alloyed powder, novel nanocomposite powder and simple mixture powder, respectively. From figure 4, it can be seen that the novel nanocomposite powder has the greatest wavelength shift and longest wavelength as compared

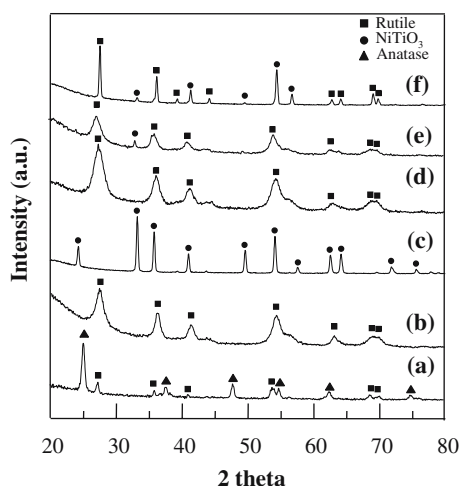


Figure 1. XRD patterns of powders (a) P-25 (b) rutile (c) NiTiO<sub>3</sub> (d) alloyed powder (e) alloyed powder after heat treatment (f) simple mixture powder (rutile + NiTiO<sub>3</sub>).

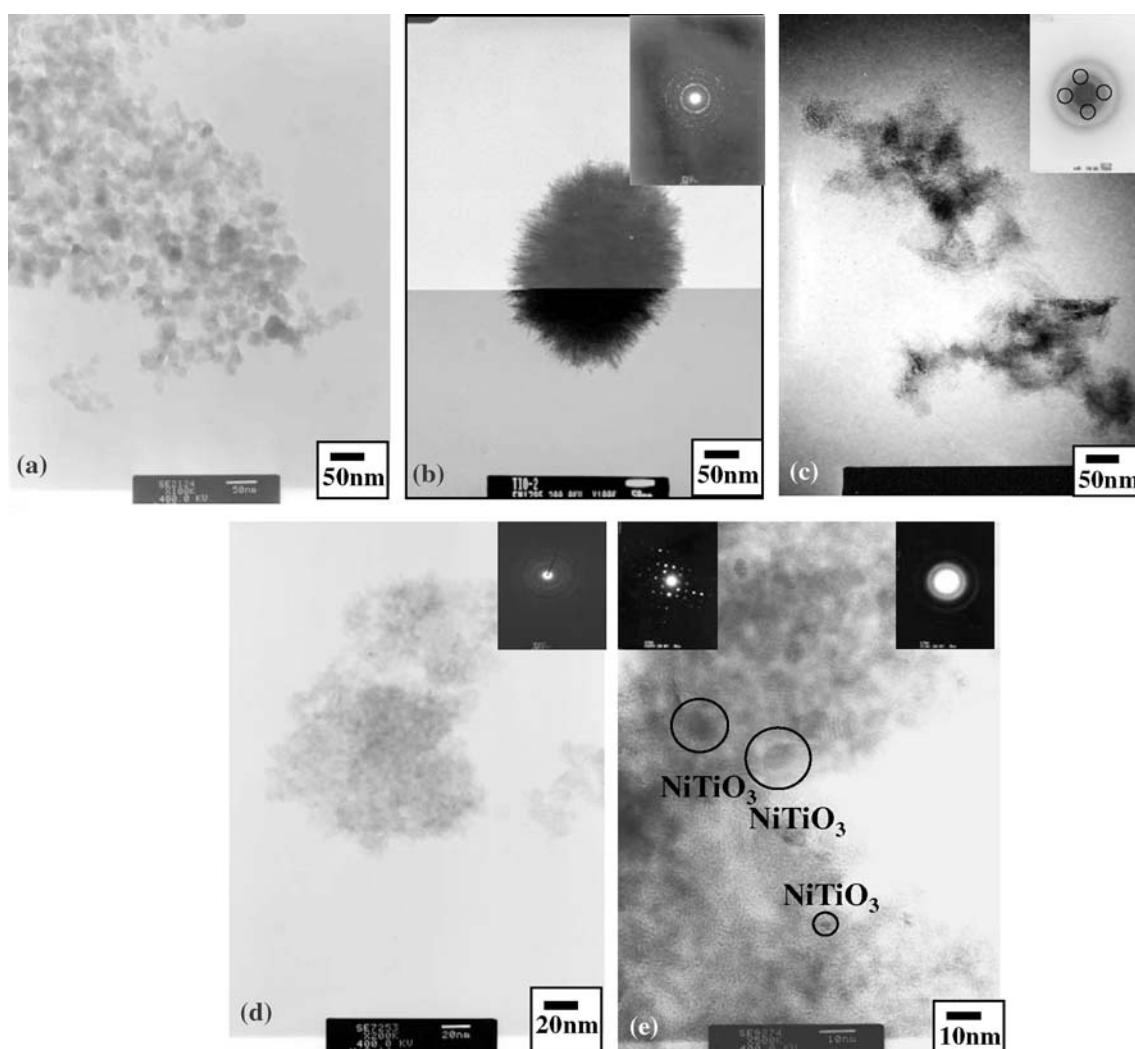


Figure 2. TEM micrographs of the powders (a) P-25 (b) rutile (c)  $\text{NiTiO}_3$  (d) alloyed powder (e) alloyed powder after heat treatment.

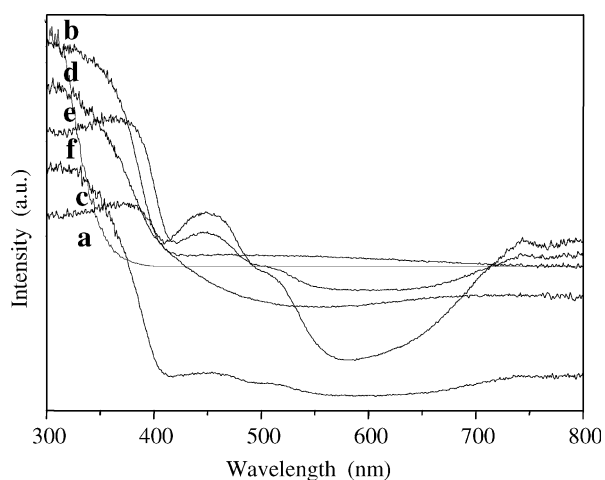


Figure 3. UV/VIS-DRS data (a) P-25 (b) rutile (c)  $\text{NiTiO}_3$  (d) alloyed powder (e) alloyed powder after heat treatment (f) simple mixture powder (rutile +  $\text{NiTiO}_3$ ).

with other powders. The most likely reason for the observed absorption shift might involve the  $\text{NiTiO}_3$  produced by thermal treatment, which can form trapping  $\text{NiTiO}_3$  band structure within the rutile band gap [16]. This trapping band structure detains the photoexcited electrons and delay the recombination rate. So, the lifetime of the electron-hole pair is extended [17]. In contrast to the well-known rutile whose absorption edge is at about 410 nm, a novel nanocomposite  $\text{TiO}_2$  exhibited obvious absorption in the visible light region over 650 nm. It indicates that even if the alloyed powder shifted the visible light region, the visible area was a little bit short and the novel nanocomposite powders ( $\text{Ni}$ -doped +  $\text{NiTiO}_3$ ) have the ability to respond to visible light. The band gap was 1.84 eV which was estimated from the absorption edge of a novel nanocomposite  $\text{TiO}_2$ . This suggests that the novel nanocomposite  $\text{TiO}_2$  has a greater ability to respond to the wavelength of the visible light region.

Figure 4 shows the photoluminescence spectra of the powders. The rutile and simple mixture powder had an

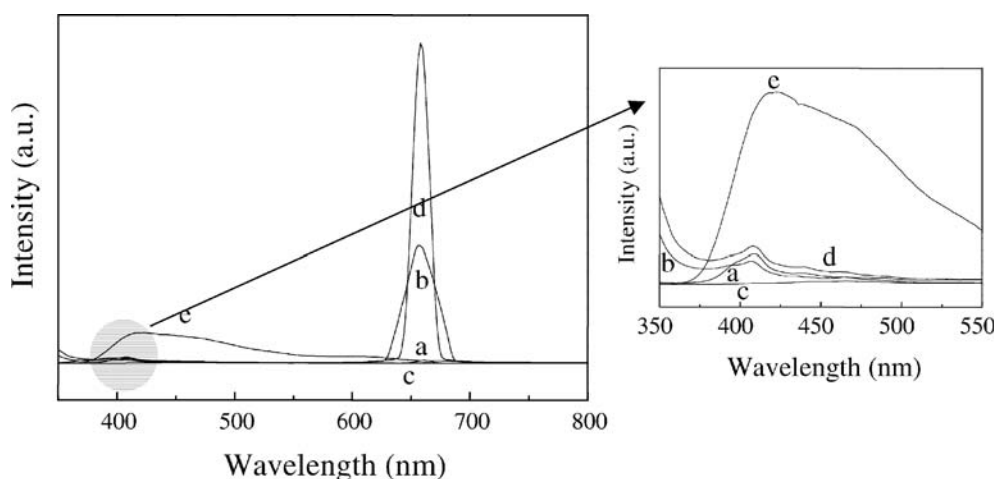


Figure 4. Photoluminescence data (a) rutile (b)  $\text{NiTiO}_3$  (c) alloyed powder (d) alloyed powder after heat treatment (e) simple mixture powder (rutile +  $\text{NiTiO}_3$ ).

emission peak at around 410 nm (3.02 eV) which corresponds to the rutile phase. And the alloyed powders showed that the emission peak shifted towards a longer wavelength region (454 nm = 2.72 eV) with decreasing peak intensity. It may be due to the fact that the new absorption was induced by localization of the trapping level in the  $\text{TiO}_2$  band gap. This result agreed well with our previous experiment [18]. However, in the novel nanocomposite powder and the  $\text{NiTiO}_3$  powder had intense emission peaks around 662 nm (1.87 eV). Also, the intensity of the novel nanocomposite powder was higher than the intensity of the  $\text{NiTiO}_3$ . This indicates that the  $\text{NiTiO}_3$  reacted in the visible light region and the novel nanocomposite powder was strongly enhanced reaction ability in visible region as compared to  $\text{NiTiO}_3$ . Because the novel nanocomposite consisted of Ni-doped  $\text{TiO}_2$  and  $\text{NiTiO}_3$  was found that the lifetime of the electrons and holes were enhanced in comparison with rutile  $\text{TiO}_2$ , Ni-doped  $\text{TiO}_2$  and single phase of  $\text{NiTiO}_3$ , as a result of electron or hole trapping at  $\text{NiTiO}_3$  band gap [19].

Photocatalytic reaction of the  $\text{TiO}_2$  powders under UV and visible light irradiation were evaluated by measuring the decomposition of 4-chlorophenol, and the results are shown in figure 5. Under UV and visible light irradiation, the reaction ability of nanocomposite  $\text{TiO}_2$  was much higher than that of P-25 and HPPLT rutile powders. And no degradation of 4-chlorophenol was observed in the absence of  $\text{TiO}_2$  powder or without irradiation. Under visible light irradiation, very small reaction of the P-25 was effect of fluorescence lamp (380–800 nm). It is very important that we synthesized a novel nanocomposite  $\text{TiO}_2$  powder which shows photocatalytic reaction under very long visible light range.

#### 4. Conclusion

In summary, the novel photocatalyst powder of  $\text{TiO}_2$  nanocomposite was synthesized by MA and heat treat-

ment. The average grain size of the particle was less than 6 nm. The composite of Ni-doped  $\text{TiO}_2$  and  $\text{NiTiO}_3$  was formed by heat treatment and dramatically enhanced the ability in the visible light reaction. The UV/VIS DRS absorption showed that the novel nanocomposite  $\text{TiO}_2$

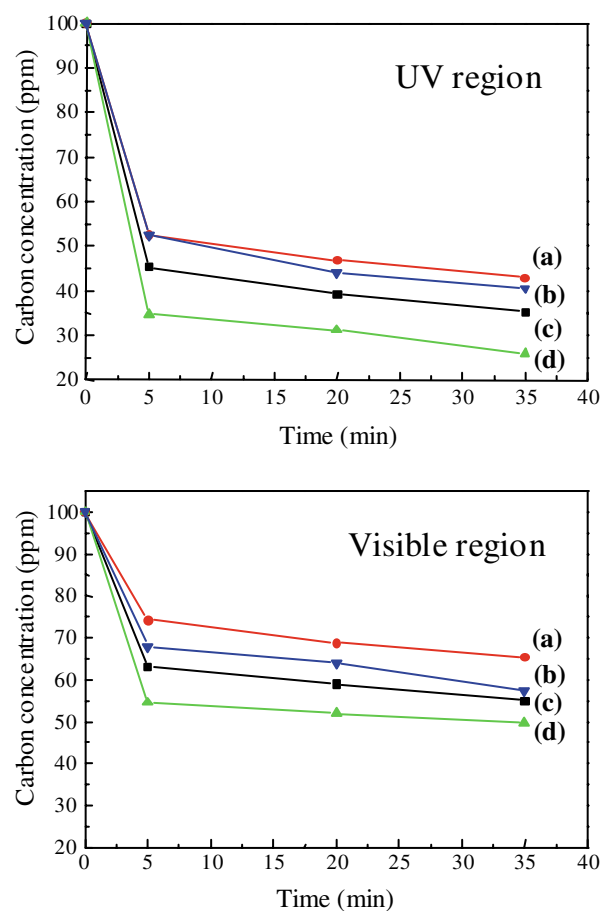


Figure 5. Photocatalytic decomposition of 4-Chlorophenol using various  $\text{TiO}_2$  powders. (a) P-25 (b)  $\text{NiTiO}_3$  (c) alloyed powder (d) alloyed powder after heat treatment.

had a higher wavelength range (650–700 nm) than Ni-doped TiO<sub>2</sub> (480–500 nm) and the rutile powder (380–400 nm). PL spectrum of the novel nanocomposite TiO<sub>2</sub> showed a new emission peak toward the longer wavelength region, suggesting a decrease in the band gap (1.87 eV).

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